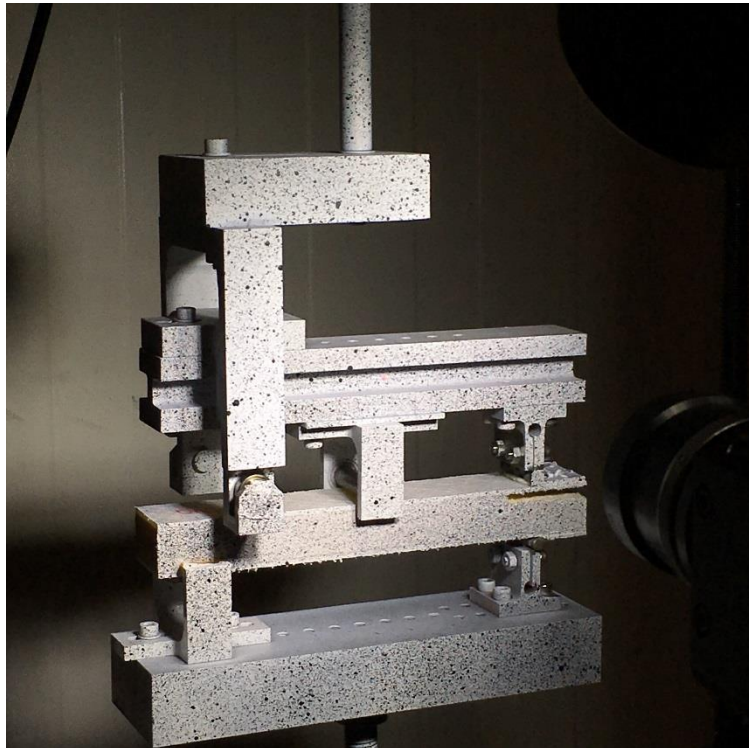


Low temperature testing of debonded PVC foam cored sandwich composites for naval vessels



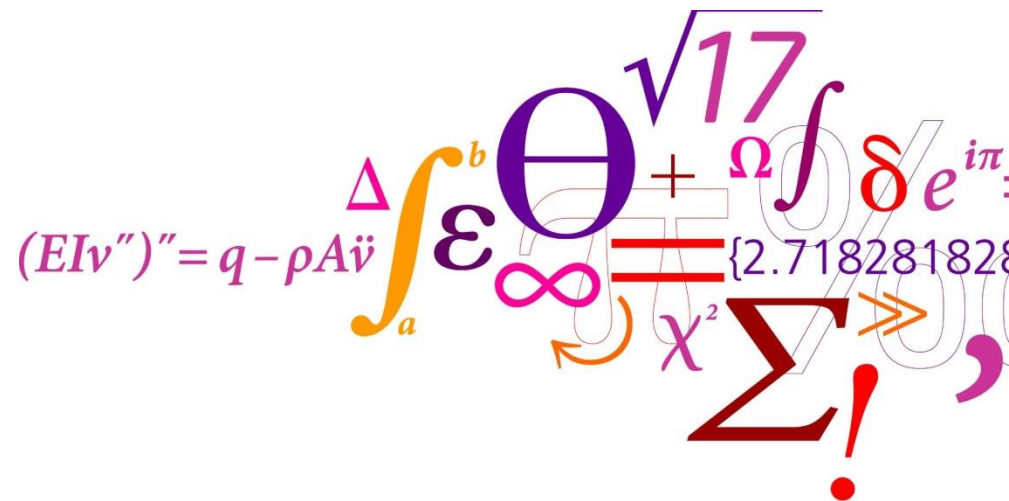
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Presenter: Arash Farshidi

PhD student

1st International Symposium on Multi-Scale Experimental Mechanics

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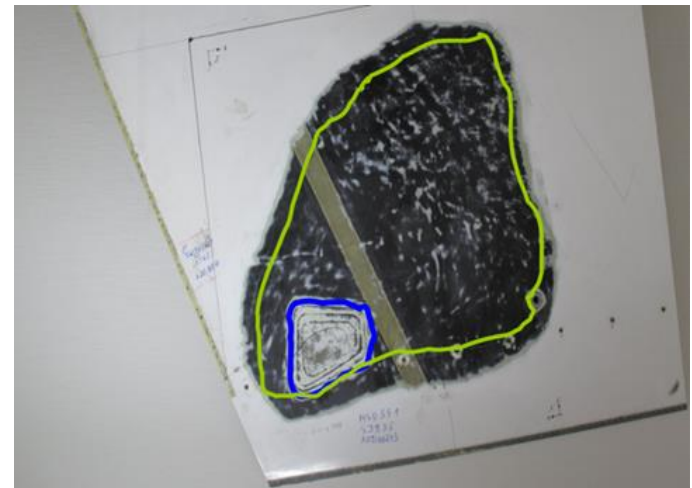


Outline

- ❑ Background and motivation
- ❑ MMB specimen
- ❑ MMB Formulation
- ❑ Implementation of tests
- ❑ G-control test methodology
- ❑ Experimental Results

Background and motivation

- + superior stiffness/weight ratio
- + Prolonged fatigue life
- + Minimization of maintenance
- + Lack of corrosion
- Experience has shown that a common problem is the existence of debond at the face/core interface of the sandwich
- Debonds can be introduced during the manufacturing processes as a result of poor resin flow or during service due to accidental overloads



Sandwich composites in naval ships

- Naval vessels are expected to operate in a variety of climatic conditions
- Arctic operations are becoming increasingly important for the Navy due to global warming
- It is important to characterize the low temperature fracture of sandwich composites



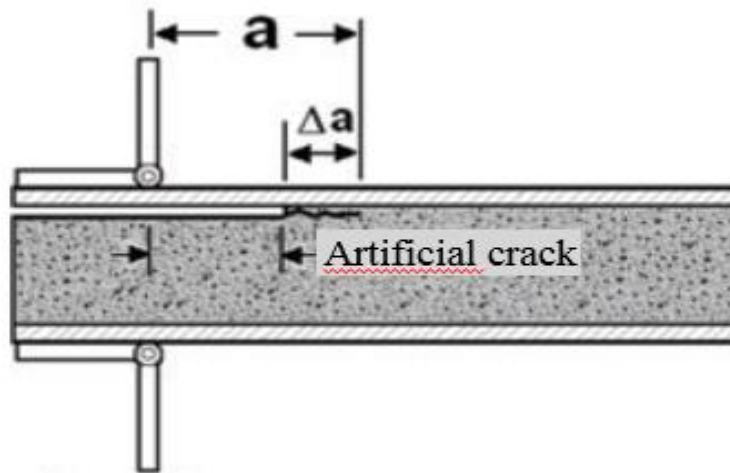
Danish naval vessel in ice filled waters

MMB Specimen

- Divinycell H100 cross-linked PVC foam
- E-glass/epoxy non-crimp multi-axial (0/45/90/-45) Devold AMT DBLT-850-E10



Mechanical properties of H100 PVC foam core and E-glass sheets.



Properties	value
Foam core: H100 PVC	
Cell size [mm]	0.45
Density [kg/m^3]	100
Compressive modulus [MPa]	135
Shear modulus [MPa]	35
Face sheet: DBLT-850 (0/45/90/-45)	
Young's modulus (E_x) [GPa]	18.6
Poisson's ratio (ν_{xy})	0.4
Shear modulus (G_{xy}) [GPa]	6.1

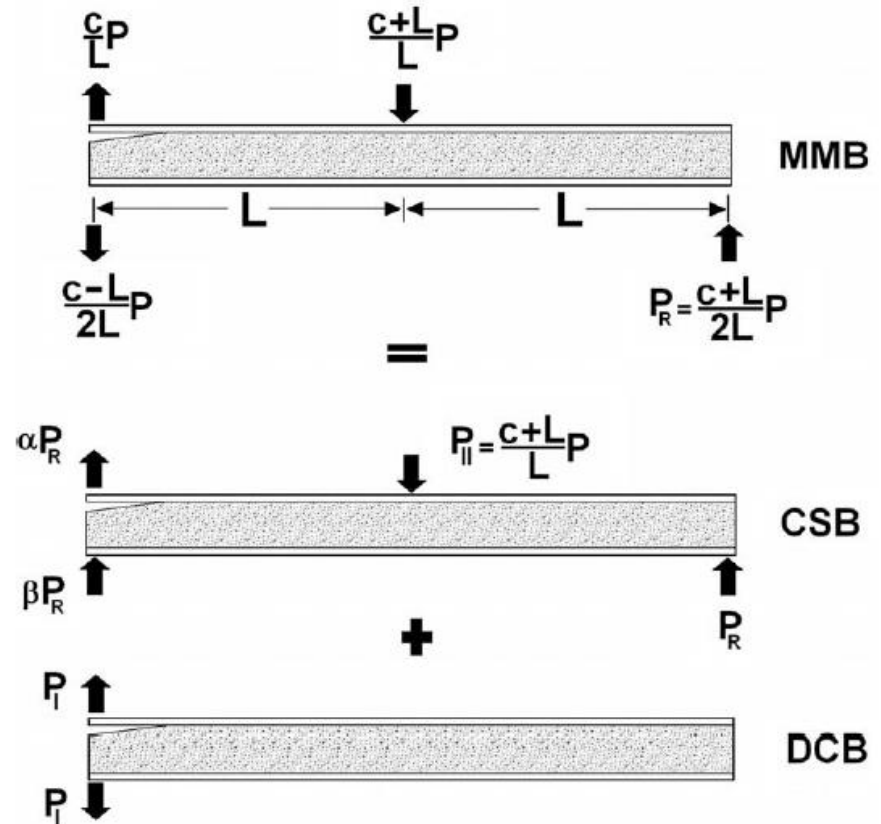
MMB formulation

$$\alpha = \left[\frac{\frac{a^3}{3} \frac{1}{D - \frac{B^2}{A}} + \frac{a}{k} \frac{1}{G_f h_f + G_{xz} h_c}}{\frac{a^3}{3} \frac{1}{D - \frac{B^2}{A}} + \frac{a}{k} \frac{1}{G_f h_f + G_{xz} h_c} + \frac{a^3}{3} \frac{1}{E_f h_f^3} + \frac{a}{k} \frac{1}{G_f h_f}} \right]$$

$$C_{DCB_lower} = \frac{a}{b} \left[\frac{1}{h_c G_{xz}} + \frac{a^2}{3(D - \frac{B^2}{A})} \right]$$

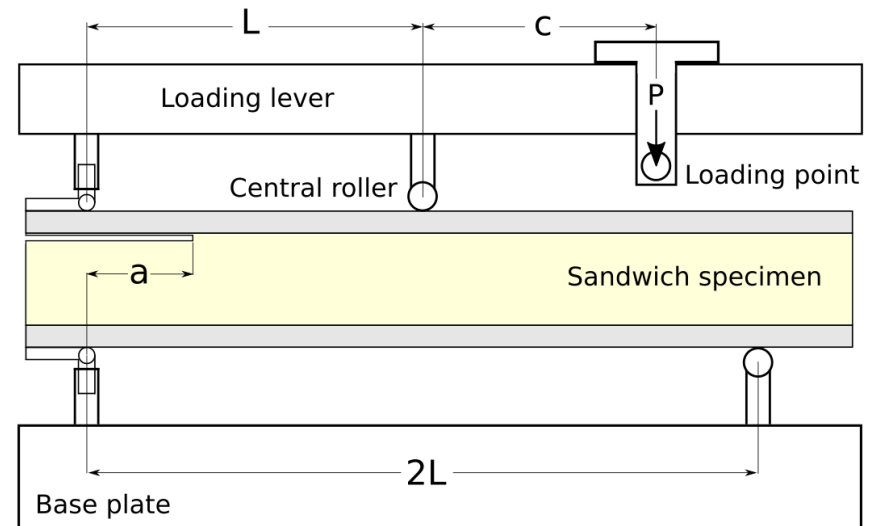
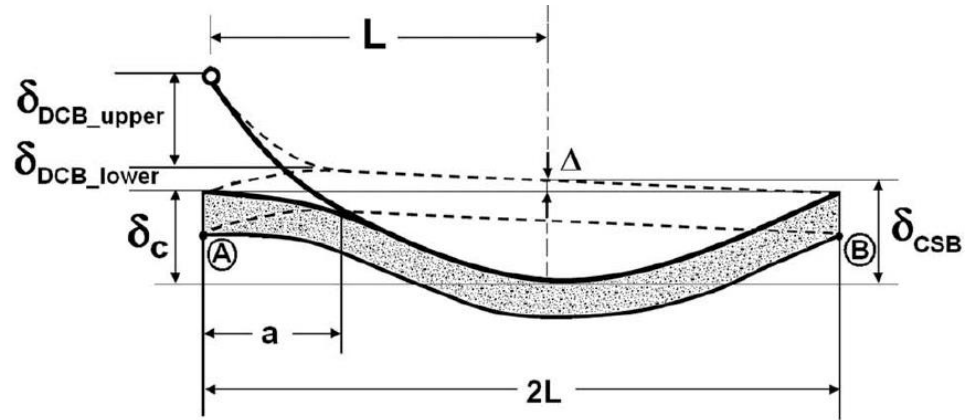
$$C_{DCB_upper} = \frac{4}{E_f h_f^3 b} \left[a^3 + 3a^2 \eta^{1/4} + 3a \eta^{1/2} + \frac{3}{2} \eta^{3/4} \right]$$

$$C_{CSB} = \frac{L^3}{6bD_{intact}} + \frac{L}{2h_c b G_{xz}} + \frac{a^3}{12b} \left[\frac{1}{D_{debonded}} - \frac{1}{D_{intact}} \right]$$



MMB formulation

- The mode-mixity phase angle has been determined using FEA and the CSDE method [C. Berggreen et al. 2007]
- The reduced mode-mixity formulation has been applied
- The mode-mixity at the crack tip is controlled by the lever arm distance, c .

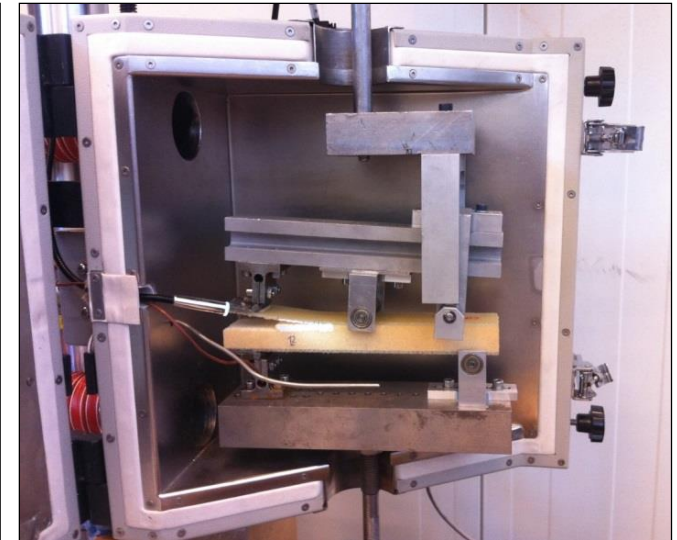


$$C = \left[\frac{c}{L} C_{DCB_upper} + \frac{c-L}{2L} C_{DCB_lower} \right] \left(\frac{c}{L} - \alpha \frac{c+L}{2L} \right) + \left(\frac{c+L}{L} \right)^2 C_{CSB}$$

$$G = \frac{P^2}{2b^2} \left(\frac{c}{L} \left(\frac{c}{L} - \alpha \frac{c+L}{2L} \right) \frac{12}{E_f h_f^3} [a^2 + 2a\eta^{1/4} + \eta^{1/2}] + \frac{c-L}{2L} \left(\frac{c}{L} - \alpha \frac{c+L}{2L} \right) \left[\frac{1}{h_c G_{xz}} + \frac{a^2}{(D - \frac{B^2}{A})} \right] + \left(\frac{c+L}{L} \right)^2 \left(\frac{a^2}{8} \left[\frac{1}{D_{debonded}} - \frac{1}{D_{intact}} \right] \right) \right)$$

Implementing MMB test inside a climatic chamber

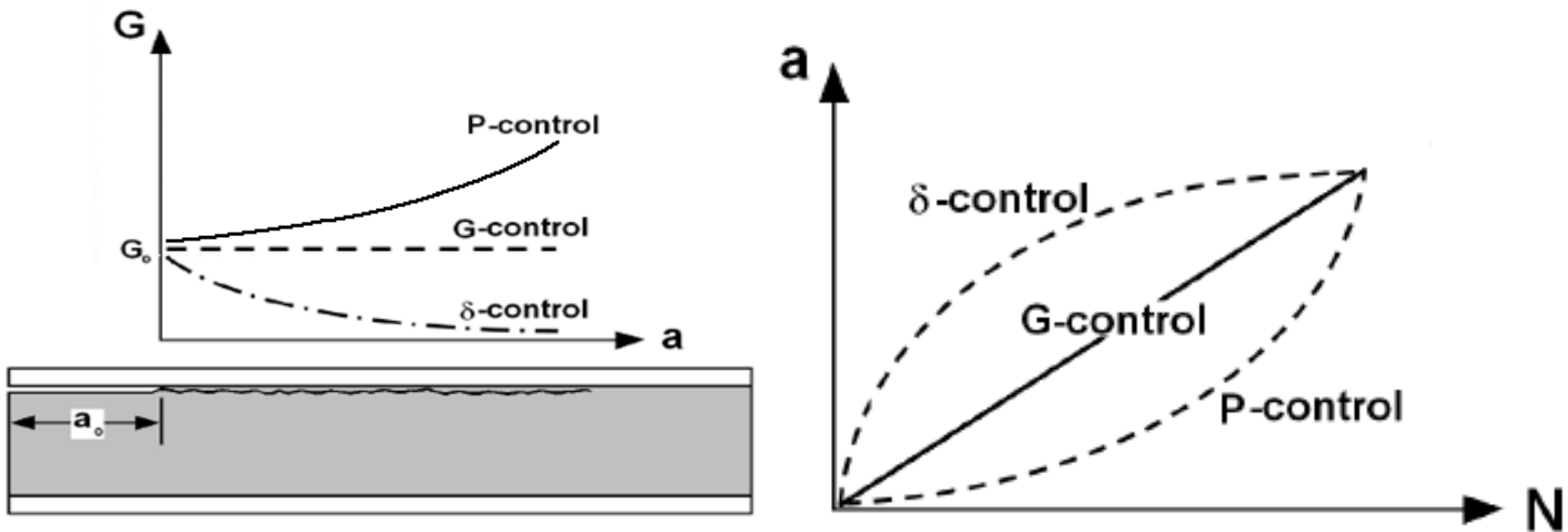
- Quasi-static and fatigue tests were carried out at mode I and mode II dominant at -20°C as well as room temperature, 23°C .



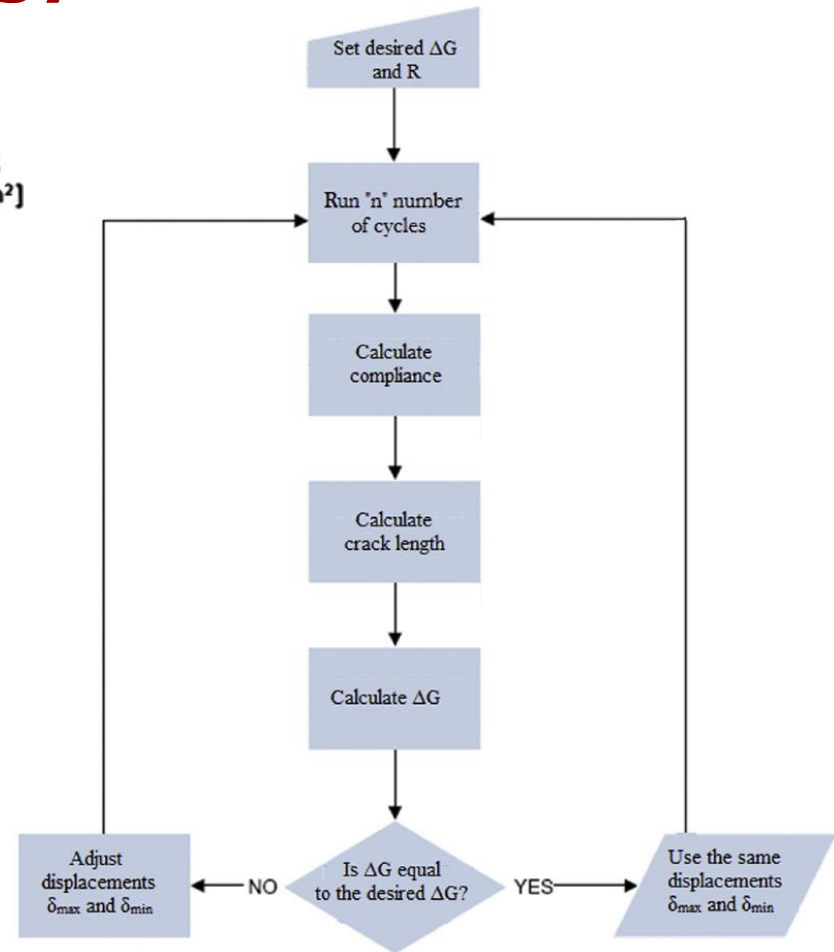
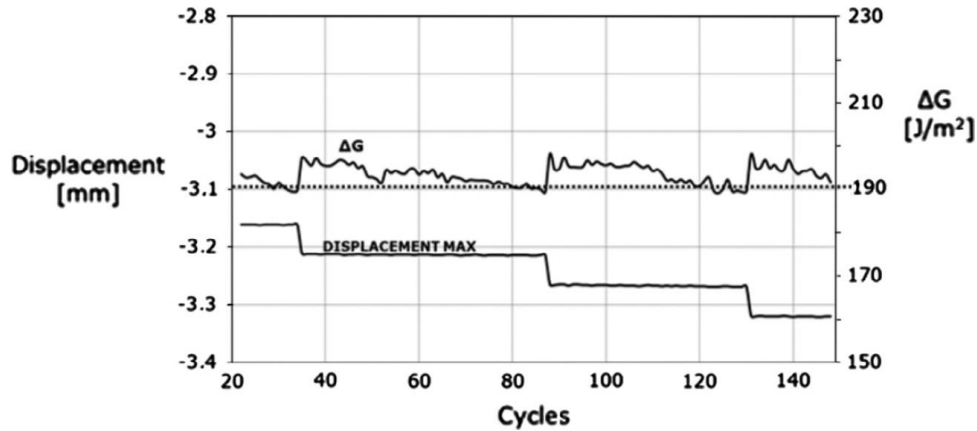
Core density	Face sheet material	Specimen length [mm]	Specimen width [mm]	Face thickness [mm]	Core thickness [mm]	Mode I ($\Psi_R=4.4^{\circ}$)	Mode II ($\Psi_R=-23.9^{\circ}$)	Low temp.	Room temp.	Fatigue	Static	No. of specimens
H100	E-glass/epoxy	220 ± 1	35 ± 0.5	2	20	Yes	No	Yes	Yes	Yes	Yes	8
H100	E-glass/epoxy	220 ± 1	35 ± 0.5	2	10	No	Yes	Yes	Yes	Yes	Yes	8

G-control test methodology

- The G-control test methodology allows highly controlled cyclic crack growth testing using real-time control of the cyclic energy release rate (ΔG).



G-control test methodology



Conclusion

The preliminary work has shown:

- It seems that the fracture toughness of a typical naval type foam core sandwich composite decreases at low temperatures.
- It seems that the intensity of the degradation depends on the mode-mixity as well.
- It seems that the stiffness properties of the foam core sandwich composite remains same at low temperatures.
- It seems that the crack growth speed increases at low temperatures.
- It seems that the increment of crack growth speed depends on the mode-mixity as well.

ACKNOWLEDGEMENT

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Thank you for your attention!

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